

図4 血管融着ハンドピース先端部

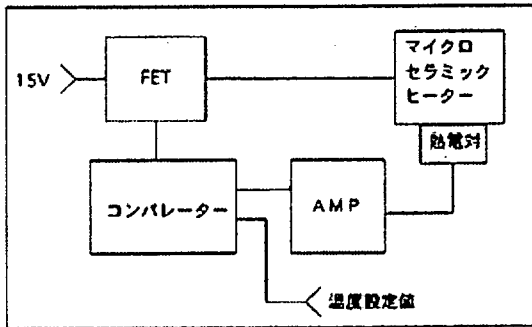


図5 温度制御方法

### B. 1. 3 ドライバユニット

血管融着ハンドピース部に内蔵された電歪素子の駆動及び温度コントロールをするための装置である。ハンドピース及びフットスイッチを接続し使用する。フットスイッチは2段式になっており、1段目でマイクロセラミックヒーターが加熱され、さらに2段目で超音波振動が開始される。振動周波数は20kHz固定であり、ボリュームにより振幅を最大8μmまでコントロール可能である。また、別のボリュームにより加熱温度を設定することが出来る。

### C. 結果

血管融着ハンドピース部の血管把持時の温度変化を測定した(図6)。血管把持により急激な温度減少が示された。設定温度になるには約100秒程度必要であった。

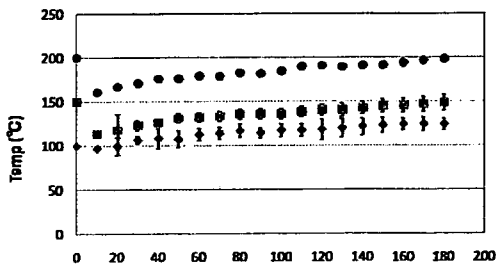


図6 超音波振動接合装置の各設定温度における

る接合試験中の温度

血管融着装置を用いて人工化合物と大動脈の接合を検討した(表1)。いずれの人工化合物においても、把持時間を延長させた場合に高強度の接合が可能であった。しかしながら、最適な把持時間は用いる人工化合物により異なった。また、把持時間が120の場合では、接合するものの組織損傷が顕著であった。以上のことから、用いる人工化合物により、最適な把持時間があることが明らかとなった。

表1 血管融着装置を用いた人工化合物と大動脈

高分子材料	振動振幅	温度(°C)	把持時間				
			10	30	60	90	120
Pellethane	MAX	200	×	②	③	□	□
Polyurethane	MAX	200	×	②	③	③	□
Vinyon	MAX	100	△	②	●	-	-
Cellophane	MAX	200	△	②	②	②	□
Polyester	MAX	200	×	①	②	②	□
PGA fiber	MAX	200	×	△	①	②	□
PGA-COL fiber	MAX	200	×	×	△	②	②

### D. 考察

血管融着デバイスの設計・製作を行った。血管融着ハンドピースの先端温度制御能および振動周波数制御能を有する。血管把持時により急激な温度低下が示され、設定温度になるには約100秒必要であった。条件を最適化することで人工化合物-大動脈の接合可能であり、接合装置としての利用可能性が示された。しかし、臨床現場で用いるためには、更なる改良が必要であると考えられる。

### E. 結論

本研究では、血管融着デバイスの設計・製作を実施し、血管融着ハンドピースの先端温度制御能および振動周波数制御能を有する超音波接合装置を開発した。人工化合物-大動脈の接合可能であったことから、接合装置としての利用可能性が示された。臨床現場で用いるためには、接合時間を短縮するため、更なる改良が必要であると考えられる。

### F. 研究発表

#### 1. 論文発表

無し

## 2. 学会発表

- 1). 濱口崇志、増澤徹、加藤綾子、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎、山本芳郎、複数低エネルギーの複合化による新しい生体接合技術の開発、生体医学第 45 巻特別号プログラム・抄録集、PS2-11-8、2007
- 2). 濱口崇志、増澤徹、加藤綾子、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎、低エネルギー複合による生体接合技術の開発、第 5 回生活支援工学系学会連合大会講演予稿集、185、2007
- 3). 河野貴宏、増澤徹、加藤綾子、濱口崇志、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎、山本芳郎、カテーテル型ステント融着マニピュレータに関する開発研究、人と福祉を支える技術フォーラム、32、2008

## 研究成果の刊行に関する一覧表

## 書籍

著者氏名	論文タイトル名	書籍全体の編集者名	書籍名	出版社名	出版地	出版年	ページ

## 雑誌

発表者氏名	論文タイトル名	発表誌名	巻号	ページ	出版年
Kwangwoo Nam, Tsuyoshi Kimura, Akio Kishida	Physical and biological properties of collagen-phospholipid polymer hybrid gels	Biomaterials	28	3153-3162	2007
Tsuyoshi Kimura, Sayaka Iwai, Toshiya Moritan, Kwangwoo Nam, Shingo Mutsuo, Hidekazu Yoshizawa, Masahiro Okada, Tsutomu Furuzono, Toshiya Fujisato, Akio Kishida	Preparation of PVA/DNA hydrogels via hydrogen bonds by ultra high pressure treatment and controlled release of DNA from hydrogels for gene delivery	Journal of Artificial Organs	10	104-108	2007
木村剛、柴久美子、加藤綾子、樋上哲哉、増澤徹、岸田晶夫	ステントグラフト内挿術での応用を目指した生体組織－高分子材料接着装置の開発	東京医科歯科大学生体材料工学研究所年報	41	11-14	2008
Yukio Ito, Tsuyoshi Kimura, Tetsuya Higami, Toshiya Fujisato, Ayako Kato, Toru Masuzawa, Akio Kishida.	Cell Culture on Nano-Vibrating Surface for Controlling Cell Function	Tissue Eng	13	1648-49	2007
Tsuyoshi Kimura, Kumiko Shiba, Kenji Yamamoto, Ayako Kato, Toru Masuzawa, Tetsuya Higami, Akio Kishida,	Polymer-biological tissue adhesion using novel pressure-vibration system	1st Asiam Biomaterials Congress Abstract		301	2007

橋本健児、増澤徹、木村孝之、加藤綾子、岸田晶夫、木村剛	ナノ振動負荷時の細胞周りの剪断応力解析	生体医工学第45巻特別号プログラム・抄録集		PS1-3-10	2007
濱口崇志、増澤徹、加藤綾子、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎、山本芳郎	複数低エネルギーの複合化による新しい生体接合技術の開発	生体医工学第45巻特別号プログラム・抄録集		PS2-11-8	2007
黒須寛秋、増澤徹、石塚健太郎、Daniel L Timms	循環系シミュレータの開発と磁気浮上型人工心臓の評価	茨城講演会講演論文集		39-40	2007
石塚健太郎、増澤徹、加藤綾子、岸田晶夫、ダニエル・ティムス	循環系治療機器の工学的評価に関する研究	第5回生活支援工学系学会連合大会講演予稿集		32	2007
橋本健児、増澤徹、木村孝之、加藤綾子、岸田晶夫、木村剛	ナノ振動刺激時における細胞膜近傍の剪断応力解析	第5回生活支援工学系学会連合大会講演予稿集		105	2007
濱口崇志、増澤徹、加藤綾子、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎	低エネルギー複合による生体接合技術の開発	第5回生活支援工学系学会連合大会講演予稿集		185	2007
増澤徹	「細胞を対象とした工学研究」の共通概念の構築—機械工学サイドから—	人と福祉を支える技術フォーラム		70	2008
小林亜美子、増澤徹、加藤綾子、木村孝之、橋本健児、岸田晶夫、木村剛	ナノ振動細胞刺激装置の開発研究	人と福祉を支える技術フォーラム		75	2008
河野貴宏、増澤徹、加藤綾子、濱口崇志、尾関和秀、木村孝之、岸田晶夫、木村剛、樋上哲哉、佐藤裕一郎、山本芳郎	カテーテル型ステント融着マニピュレータに関する開発研究	人と福祉を支える技術フォーラム		32	2008
永島勲、尾関和秀	生体材料を目的としたDLCコーティング高分子材料の機械的特性評価	平成19年度茨城講演会予稿集		53-54	2007
吉原裕貴、尾関和秀	スパッタリング法により作製されたハイドロキシアパタイト薄膜の薬剤吸着特性評価	平成19年度茨城講演会予稿集		55-56	2007

永島勲、佐藤修平、尾関和秀	生体材料を目的としたDLCコーティング高分子材料の機械的特性評価	第23 回ライフサポート学会予稿集		109	2007
吉原裕貴、尾関和秀	スパッタリング法によるハイドロキシアパタイト薄膜の薬剤吸着特性評価	第23 回ライフサポート学会予稿集		110	2007
吉原裕貴、尾関和秀、増澤徹	スパッタHA 薄膜の薬剤吸着特性	第20 回アパタイト研究会予稿集		42-43	2007
石沢泰輔、木村孝之、増澤徹	二次元集積化磁気センサの応答速度の検討	電気学会東京支部茨城支所研究発表会		PB13	2007



## Physical and biological properties of collagen-phospholipid polymer hybrid gels

Kwangwoo Nam, Tsuyoshi Kimura, Akio Kishida\*

*Division of Biofunctional Molecules, Institute of Biomaterials and Bioengineering, Tokyo Medical and Dental University, 2-3-10 Kanda-Surugadai, Chiyoda-ku, Tokyo, 101-0062, Japan*

Received 6 December 2006; accepted 8 March 2007

Available online 14 March 2007

### Abstract

We successfully developed a novel method for immobilizing poly(2-methacryloyloxyethyl phosphorylcholine) [Poly(MPC)] polymer onto collagen using *N*-(3-dimethylaminopropyl)-*N'*-ethylcarbodiimide (EDC) and *N*-hydroxysuccinimide (NHS) as cross-linkers. In order to obtain the highest possible molar ratio of immobilized MPC moieties on the collagen gel, a collagen-phospholipid polymer hybrid gel was prepared by repeating the cross-linking process up to three times to create a dense network of collagen and PMA. Network formation by repeating the immobilization process was successful, resulting in decreased free amine group content and a low swelling ratio. The hybrid gel displayed very high stability against degradation by collagenase and possessed high hydrophilicity. Fibrinogen adsorption and cell adhesion were reduced and demonstrated less cell proliferation as compared to that by uncross-linked collagen gel. The collagen-phospholipid polymer hybrid gel did not exhibit toxicity, and the cell morphology remained intact (round); this implies that the interaction between the cell and the collagen-phospholipid polymer hybrid gel is safe and mild.

© 2007 Elsevier Ltd. All rights reserved.

**Keywords:** Collagen; Phospholipid polymer; Immobilization; Protein adsorption; Cell adhesion

### 1. Introduction

In order to use collagen as a biomaterial product, cross-linking of collagen and/or immobilizing synthetic polymers onto collagen are indispensable measures. Non-treated natural collagen cannot be directly applied to a biological system due to drawbacks such as poor mechanical strength, calcium deposition, and high thrombogenicity. However, collagen is biocompatible, non-antigenic, synergic with bioactive components, easily modifiable, and abundantly available; these attributes render it suitable for medical application [1]. Hence, the undesirable properties of collagen should be eliminated while retaining its desirable properties.

When cross-linking the collagen gel, it should be ensured that the cross-linker is not toxic and does not affect biocompatibility. Preparing a cross-linked collagen gel does not necessarily require chemical cross-linking. Diverse

methods such as chemical and physical cross-linking, UV irradiation, and blending have been used to cross-link collagen [2–6]. Among these, cross-linking using *N*-(3-dimethylaminopropyl)-*N'*-ethylcarbodiimide (EDC) and *N*-hydroxysuccinimide (NHS) was chosen for this study [7–9]. Cross-linked collagen with EDC and NHS results in “zero-length” amide cross-links between the carboxylic acid groups from aspartic and glutamic acid residues, and the  $\epsilon$ -amino groups from (hydroxy-)lysine residues [9]; these form intra- and interhelical cross-links to provide an EDC/NHS cross-linked collagen gel. A 2-methacryloyloxyethyl phosphorylcholine (MPC)-based copolymer, namely, poly(MPC-*co*-methacrylic acid) (PMA), which is also a well-known hemocompatible material [10], was used to cross-link the microfibrils of collagen to produce a collagen-polymer hybrid gel [11].

In our previous study, we discovered that the collagen-polymer hybrid gel could be prepared efficiently under alkaline pH conditions. Immobilization of PMA onto collagen would cover the entire collagen surface, increase the mechanical strength, reduce water absorption, and

\*Corresponding author. Tel.: +81 3 5841 8028; fax: +81 3 5841 8005.

E-mail address: [Kishida.fm@tmd.ac.jp](mailto:Kishida.fm@tmd.ac.jp) (A. Kishida).

impart durability against collagenase degradation. However, a high percentage of MPC did not immobilize on the collagen gel surface. A longer coupling time or use of larger amount of EDC, NHS, or PMA did not result in an increase in the amount of immobilized PMA. This is believed to be attributable to spatial problems. For immobilization of heparin, low adoption of the polymer with the EDC/NHS coupling reaction was considered a problem. With regard to heparin, it is known that approximately 5.5% immobilization can be achieved [12]. However, activation of the carboxylic group can be triggered at any time point [7]. The coupling reaction continues when EDC/NHS is re-added to collagen gel. Based on this, we developed a novel technique to immobilize the polymer—the “repeat immobilization process.”

The repeat immobilization process comprises simply repeating the process of immobilization on the surface of the pre-activated polymer, and a polymer-immobilized collagen gel is thus obtained. The rationale for this process is based on the following: (1) the carboxyl groups can be activated at any time point, and (2) a high percentage of unreacted amine groups are available. Re-activation may promote formation of additional amide bonds, which could not be achieved by a longer coupling time or the use of a higher amount of EDC, NHS, or PMA. In this study, we repeated the immobilization process until a collagen gel with the highest possible number of phospholipid head groups was obtained. Using this gel, we characterized the physical and biological properties of the collagen-polymer hybrid gel. To distinguish the collagen-polymer hybrid gel prepared by the repeat immobilization process from that prepared by time control, we named the former as “collagen-phospholipid polymer hybrid” (a CoPho gel).

## 2. Experimental methods

### 2.1. Preparation

#### 2.1.1. Synthesis of PMA

PMA was synthesized according to a previously published method [11,13]. In brief, MPC and methacrylic acid (MA) were dissolved in an ethanol solution. Subsequently, a specific amount of 2,2-azobisisobutyronitrile (AIBN) was added to the ethanol solution. Polymerization was performed in a completely sealed round-bottom flask for 16 h at 60 °C. The solution was precipitated into diethyl ether, freeze-dried, and stored in vacuo until further use. The molar ratio of PMA was MPC:MA = 3:7, and the average molecular weight was  $3 \times 10^5$ .

#### 2.1.2. Preparation of EDC and NHS cross-linked collagen gel (EN gel)

Cross-linked collagen gel was prepared by a previously reported method [11]. Instead of the 0.5 wt% collagen type I solution (pH 3; KOKEN, Tokyo, Japan), 2 wt% collagen type I solution was prepared and used for the film preparation. The collagen solution was dropped onto the polyethylene film and allowed to dry at room temperature. The collagen film (thickness =  $56 \pm 3 \mu\text{m}$ ) was immersed in a 0.05 M 2-morpholinoethane sulfonic (MES) acid buffer (pH 9.0) (Sigma, St Louis, USA) containing EDC (Kanto Chemicals, Tokyo, Japan) and NHS (Kanto Chemicals). The molar ratio of the constituents was EDC:NHS:collagen-carboxylic acid groups = 10:10:1. The cross-linking procedure was allowed to proceed at 4 °C for 4 h to produce a cross-linked gel (EN

gel). After 24 h, the reaction was terminated by removing the gel from the solution. Subsequently, the gel was first washed with a 4 M aqueous  $\text{Na}_2\text{HPO}_4$  solution for 2 h to hydrolyze any remaining O-acylisourea groups and subsequently with distilled water over a duration of 3 days to remove traces of salts from the gel.

#### 2.1.3. Preparation of MPC-immobilized gel (MiC gel, MdC gel, and MtC gel; CoPho gel)

MPC-immobilized collagen gel (MiC gel) was prepared using uncross-linked collagen gel (immersed in an alkaline solution at pH 9.0 for 30 min) or EN gel [11]. PMA was added to the MES buffer (pH 9.0) along with EDC and NHS and was activated for 10 min before the uncross-linked collagen or EN gel was immersed. The molar ratios of each chemical was fixed; EDC:NHS:collagen-carboxylic acid groups = 10:10:1. The immobilization of PMA to collagen continued for 48 h at 4 °C. Subsequently, the gel was first washed with 4 M aqueous  $\text{Na}_2\text{HPO}_4$  solution for 2 h and then with distilled water for 1 day to remove traces of salts and thus prepare a salt-free MiC gel. Fig. 1 illustrates the basic scheme for activation of PMA by EDC and NHS immobilization on collagen. To increase the number of MPC moieties on the collagen-polymer hybrid gel, a MPC-double immobilized collagen (MdC) gel was prepared by immobilizing PMA on the MiC gels by using the same procedure as earlier. To investigate the possibility of further immobilization, we prepared a MPC-triple immobilized collagen (MtC) gel using MdC as the base collagen for PMA immobilization. MiC-0, MdC-0, and MtC-0 were prepared from uncross-linked collagen gels; MiC-1, MdC-1, and

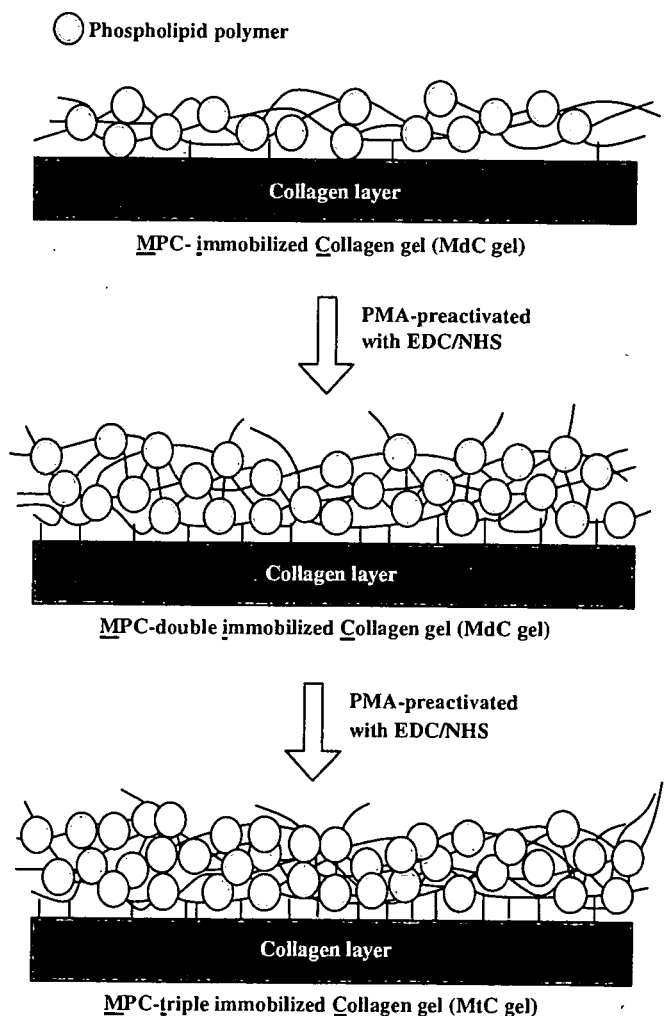


Fig. 1. Schematic diagram of the immobilization process of PMA on collagen.

MtC-1, from EN gels. The terminology used for the gel samples is listed in Table 1.

#### 2.1.4. Preparation of glutaraldehyde cross-linked collagen gel (G-gel)

Cross-linking collagen gel to glutaraldehyde was performed by a previously reported method [14]. In brief, a 25% glutaraldehyde solution (Merck, Darmstadt, Germany) was diluted to 0.5 wt% in phosphate buffer solution (PBS). The collagen film was immersed in the glutaraldehyde/PBS solution and was cross-linked for 3 h at room temperature. After cross-linking, the sample was first rinsed in running tap water for 30 min and then in 4 M NaCl for 2 h. In order to eliminate NaCl, the sample was rinsed with distilled water for 1 day to yield a glutaraldehyde cross-linked collagen gel (G-gel). The physical and biological properties of this gel were compared with those of the CoPho gels.

## 2.2. Characterization

### 2.2.1. Surface analysis

Surface analysis was executed using X-ray photoelectron spectroscopy (XPS; AXIS-HSi, Shimadzu/KRATOS, Kyoto, Japan) and static contact angle (SCA; ERMA-G1, Tokyo, Japan). The samples that had been cut into small pieces were lyophilized overnight. The chemical composition of the gel surface was determined by the releasing angle of the photoelectrons fixed at 90°. SCA measurement was performed by using a contact angle goniometer (ERMA-G1, Tokyo, Japan) and a Bil-mont syringe. The contact angle of the drop on the surface was measured at room temperature. The SCA experiment was repeated 7 times, and the average was calculated together with the standard deviation.

### 2.2.2. Determination of the reacted amine group content

The concentration of the primary amine group in tissue samples was determined using a colorimetric assay [15,16]. Three to four milligrams of each sample was prepared. These samples were placed in a 4 wt% aqueous NaHCO<sub>3</sub> solution (Kanto Chemicals, Tokyo, Japan) and 2,4,6 trinitrobenzene sulfonic (TNBS) acid. Subsequently, 0.5 wt% aqueous TNBS solution was added (Wako chemicals, Osaka, Japan). The reaction was allowed to proceed for 2 h at 40 °C; the samples were rinsed with saline solution in a vortex mixer to remove unreacted TNBS. Subsequent to freeze-drying the samples overnight, the dry mass was determined. The dry samples were immersed in 2 mL of 6 M aqueous HCl until fully dissolved. The resultant solution was subsequently diluted with distilled water (8 mL) and absorbance was measured at 345 nm (V-560, Jasco, Tokyo, Japan). The concentration of reacted amine groups was calculated using the following equation [16]:

$$[\text{NH}_2] = \frac{A \times V}{\epsilon \times l \times m}, \quad (1)$$

where [NH<sub>2</sub>] denotes the reacted amine group content (mol/g collagen gel),  $\epsilon$  the molar absorption coefficient of trinitrophenyllysine ( $1.46 \times 10^4$  mL/mmol cm),  $A$  the absorbance,  $V$  the volume of the solution (mL),  $l$  the path length (cm), and  $m$  the weight of the sample (mg). The reacted amine group contents of respective collagen gels were all compared with Uc gel.

Table 1  
Terminology of collagen gels used in this study

Terminology	Composition
Uc-gel	Uncross-linked gel (stabilized under pH 9.0)
EN gel	EDC/NHS-cross-linked collagen gel under pH 9.0
CoPho gel	
MiC-0	PMA immobilized on Uc-gel under pH 9.0
MdC-0	PMA immobilized on MiC-0 gel under pH 9.0
MtC-0	PMA immobilized on MdC-0 gel under pH 9.0
MiC-1	PMA immobilized on EN gel under pH 9.0
MdC-1	PMA immobilized on MiC gel under pH 9.0
MtC-1	PMA immobilized on MdC gel under pH 9.0

### 2.2.3. Swelling test

The swelling test of the samples was executed by cutting the lyophilized gels into small pieces and placing them in a neutral pH aqueous solution at 37 °C. The pH of the aqueous solution was adjusted to 7.4. The gels were gently shaken for 24 h and were measured for assessing the change in weight of the sample. Swelling ratio was calculated in order to define the swelling phenomenon accomplished by water absorption. The experiment was repeated 5 times and the average was calculated along with standard deviation. The following equation was used to calculate the swelling ratio.

$$\text{Swelling ratio, } S(\%) = \frac{W_h - W_d}{W_d} \times 100,$$

where  $W_h$  denotes hydrated weight of the gel and  $W_d$  the dry weight of the gel.

### 2.2.4. Fibrinogen adsorption test

Bioresponse was evaluated in terms of protein adsorption by using bovine plasma fibrinogen. The concentration was adjusted to 1 mg/mL. First, the collagen gels were equilibrated by immersing them in PBS. Subsequently, the gels were transferred to the fibrinogen solution, and the solution was incubated for 3 h. After rinsing with PBS, the adsorbed fibrinogen was recovered by dipping the samples in 1 wt% *n*-sodium dodecyl sulfate (SDS) for 60 min [17]. The concentration of recovered fibrinogen was determined at 490 nm by using a Micro BCA kit (Bio-rad, Model 680, Tokyo, Japan).

### 2.2.5. Cell adhesion test

The interaction between the L929 cells (mouse fibroblast) and the collagen gels was evaluated. The fibroblasts were cultured in Eagle's Minimum Essential Medium (E-MEM, Gibco, NY, USA) supplemented with 10% fetal bovine serum (FBS, Gibco, NY, USA) at 37 °C in 5% CO<sub>2</sub> atmosphere. After treatment with 0.25% trypsin, the cell density was adjusted to  $5 \times 10^3$  cells/mL and the cells were seeded on the gel surface [18,19]. The collagen gels were sterilized by placing the gels first in an ethanol:water (50:50) solution for 2 h, then in a 70:30 solution for 2 h, and overnight in a 100:0 solution before lyophilization. The lyophilized gels were hydrolyzed with E-MEM for 5 min, and the E-MEM was disposed immediately prior to cell seeding. After 24- and 48-h cycles, the number of adhering cells was measured using UV-vis spectrophotometer (V-560, Jasco, Tokyo, Japan) at 560 nm by lactate dehydrogenase (LDH) assay. All experiments were repeated 3 times and the average was calculated together with the standard deviation.

The morphology of the L929 cells after the 48-h incubation period was observed using scanning electron microscopy (SEM). The cells attached to the samples were rinsed with PBS and fixed with 2.5% glutaraldehyde. Subsequently, the samples were dehydrated for 15 min using an ethanol dilution series (10%, 30%, 50%, 70%, and 90%) before the final dehydration with 100% ethanol. After dehydration, the samples were first dried at room temperature and then in vacuum. In order to avoid deformation of the cells, all samples were fixed onto the glass cover prior to vacuum drying.



### 2.2.6. Cell viability test

The cell viability test was executed using the 3-(4,5-dimethylthiazolyl)-2,5-diphenyltetrazolium bromide (MTT, Sigma Chemical Co., St. Louis, USA) assay. The cells were cultured on each test specimen (5000 cells/well) for 48 h and were washed twice with PBS. Subsequently, 200  $\mu$ l of the MTT solution (0.5 mg/mL in medium, filter sterilized) was added to the culture wells. After incubation for 4 h at 37 °C in a 10% CO<sub>2</sub> atmosphere, the MTT reaction medium was removed and blue formazan was solubilized by the addition of 100  $\mu$ L dimethylsulfoxide (DMSO). The optical density readings were subsequently performed at 570 nm by using the Micro BCA kit.

### 2.2.7. Statistical analysis

All the experiments were repeated at least 3 times and the values were expressed as mean  $\pm$  standard deviation. In several figures, the error bars are not visible because they are included in the plot. Statistical analysis was performed using Student's *t*-test with the significant level set as  $P < 0.05$ .

## 3. Results

Atomic concentration percentage of phosphorus is shown in Table 2. Phosphorus was not detected in the case of Uc gel and EN gel, while phosphorus was detected in the case of CoPho gels. The phosphorus concentration increased for MdC gels, but significant increase in the phosphorus atomic concentration was not shown for MtC gels.

SCA of the respective collagen gels was measured and is illustrated in Fig. 2. The SCA for Uc and EN gels was approximately 70°. SCA decreases as a result of repeating the immobilization process, and was 20° for MtC-0 and -1 gels, thus indicating that the CoPho gel was acquiring a hydrophilic nature.

Fig. 3 illustrates the results of the reacted amine group content of the respective collagen gels. It decreases from approximately 60–30% as the PMA is immobilized compared to Uc gel. The lowest reacted amine group content was observed for G-gel, which was approximately 15% of Uc gel.

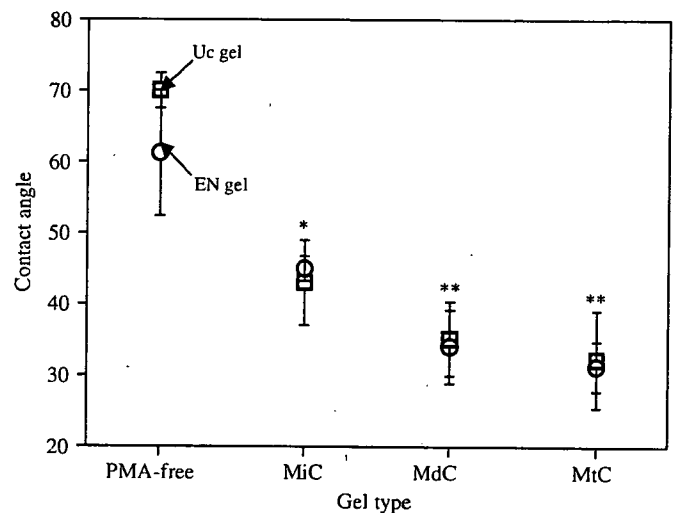
Fig. 4 illustrates the swelling ratio of the collagen gels under acidic and neutral pH conditions. Uc gel dissolved in acidic pH conditions (pH 2.4) and swelled to approximately 350% in neutral pH conditions (pH 7.4). The swelling ratio decreased rapidly on execution of cross-linking. In terms of the swelling ratio, the two gels did not

differ substantially under neutral pH conditions, contrary to the situation under acidic pH conditions. However, a decrease in the swelling ratio was observed; after re-immobilization it decreased from 130% for MiC-1 gel to 95% for MtC-1 gel.

Degradation by collagenase demonstrated that cross-linking decreases the degradation rate of collagen gels (Fig. 5). Uc gels, which completely degraded within 6 h, remained undegraded for 24 h when cross-linked with EDC/NHS alone. Immobilization stabilized the gels against degradation by collagenase.

Fig. 6 illustrates the results of fibrinogen adsorbed on the surface of the collagen gel. It can be clearly seen that fibrinogen adsorption decreased as cross-linking proceeded. The amount of adsorbed fibrinogen further decreases as the MPC polymer is immobilized.

Fig. 7 illustrates the results of the cell adhesion test. Here, we discovered that repeated immobilization of PMA suppressed cell adhesion. Comparison of cell adhesion on completion of 24- and 48-h cycles revealed that the number of adhered cells for in the case of the Uc gel after 48-h cycle had increased by approximately 2 times; the rate of



Uc gel and EN gel  $p < 0.01$  vs. \* and \*\*  
\*  $p < 0.01$  vs. \*\*

Fig. 2. The static contact angle of the collagen gels. Each value represents the mean  $\pm$  SD ( $n = 5$ ).

Table 2  
Atomic phosphorus concentration of respective collagen gels

Gel type	Atomic phosphorus concentration (%)
Uc gel	0
EN gel	0
CoPho gel	
MiC-0	0.21 $\pm$ 0.06
MdC-0	0.36 $\pm$ 0.05
MtC-0	0.37 $\pm$ 0.06
MiC-1	0.18 $\pm$ 0.02
MdC-1	0.37 $\pm$ 0.09
MtC-1	0.36 $\pm$ 0.05

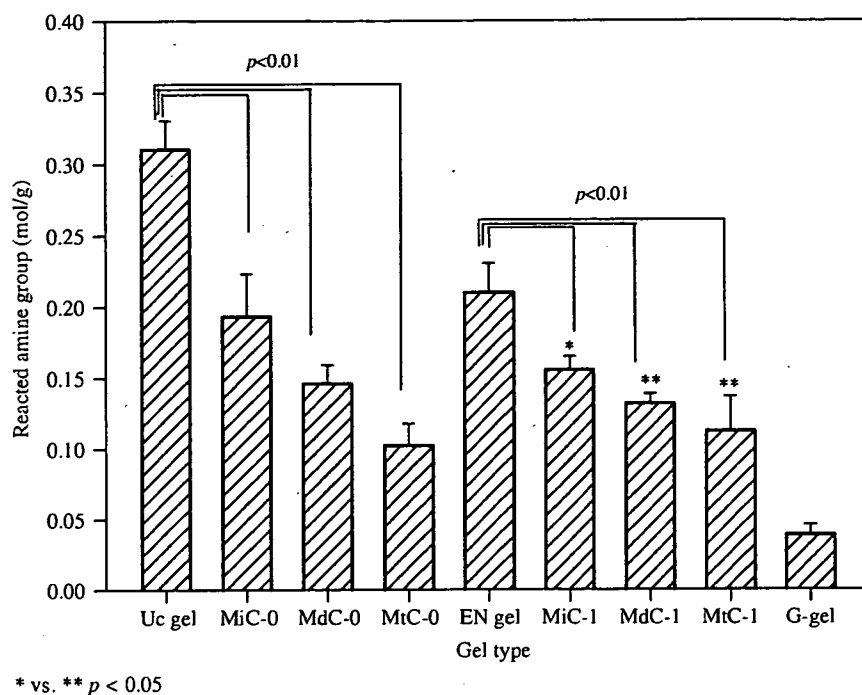


Fig. 3. The percentage of free amine group contents of respective collagen gels. Each value represents the mean  $\pm$  SD ( $n = 5$ ).

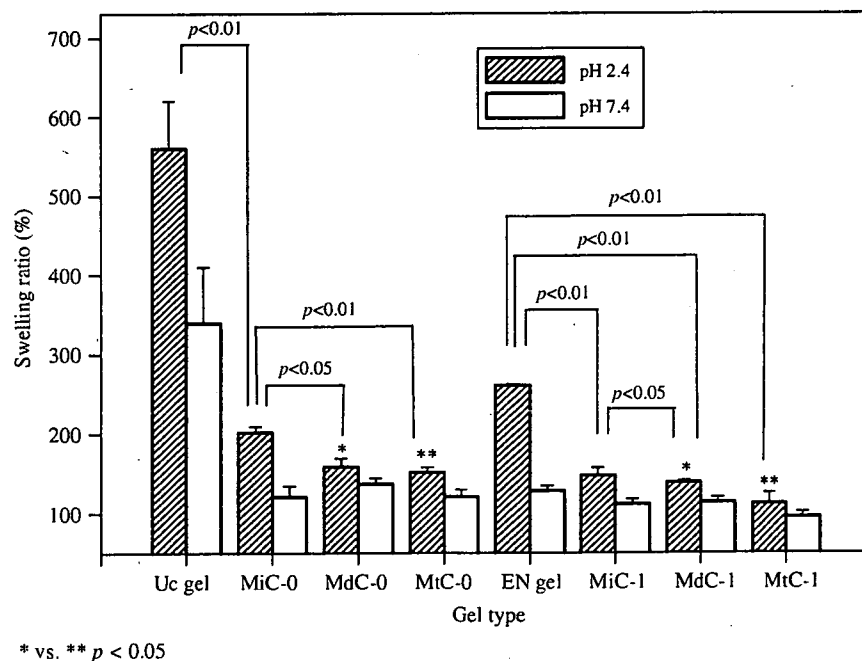


Fig. 4. Swelling ratio of the collagen gels under acidic pH conditions (pH 2.1) (hatched bar) and under neutral pH conditions (pH 7.4) (empty bar). Each value represents the mean  $\pm$  SD ( $n = 5$ ).

increase decreases as the cross-linking of the collagen gels is proceeded. When assessing the suppression of cell adhesion in terms of intra- and interhelical cross-links, we observed that higher suppression was considerably higher in gels with intra- and interhelical cross-link. MtC-1 gel displayed cell adhesivity that was similar to that of the G-gel.

Cell morphology observed using SEM (Fig. 8) demonstrated that the L929 cells were deformed on the non-MPC surface. On the other hand, the cells remained intact (round) on the CoPho gel surface. An increase in the density of the MPC head group resulted in a decrease in the distribution of L929 cells.

Fig. 9 illustrates the viability of the L929 cells after 48 h. It reveals that cell viability ranges from 96% to 115% (TCPS as 100%) [20]. Cell viability was approximately 70% for the glutaraldehyde cross-linked collagen gel. Immobilization of PMA did not induce any toxicity.

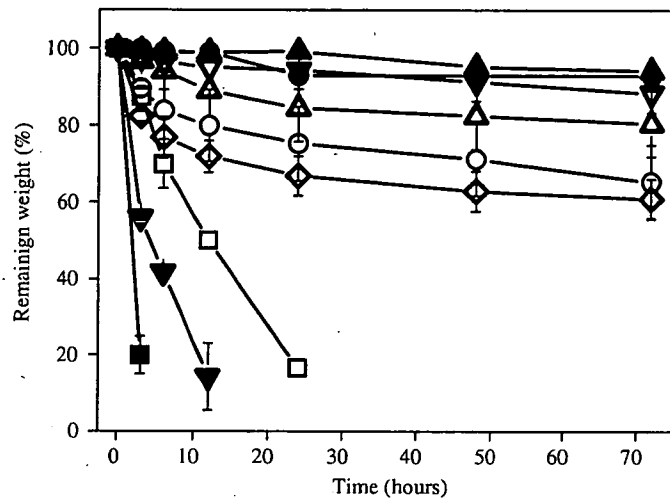


Fig. 5. Degradation of collagen gels by collagenase in Tris-HCl buffer (pH 7.4) at 37 °C. (■) Uc gel, (●) MiC-0 gel, (▲) MdC-0 gel, (▼) MtC-0 gel, (□) EN gel, (○) MiC-1 gel, (△) MdC-1 gel, (▽) MtC-1 gel, and (◇) G-gel. Closed symbols indicate gels without interhelical cross-links while open symbols indicate gels with interhelical cross-links. Each value represents the mean  $\pm$  SD ( $n = 5$ ).

## 4. Discussion

### 4.1. Physical properties of the EN and CoPho gels

The collagen gel that was prepared from a 2 wt% aqueous collagen solution differed from that prepared from a 0.5 wt% collagen solution [11]. A considerably thicker film was obtained ( $\approx 50 \mu\text{m}$ ), and this film displayed tougher mechanical strength, suppressed swelling, and it slowed collagenase degradation. However, thermodynamic conditions such as shrinkage temperature remained unaltered.

XPS signals displayed a phosphorus peak and a nitrogen peak [ $\text{N}^+(\text{CH}_3)_3$ ] at 134 and 403.2 eV, respectively; this indicates that PMA was effectively adopted [10,11]. This implies that PMA was successfully immobilized on the surface of the collagen gels. The phosphorus concentration would increase when the MPC is immobilized on the collagen, but did not increase significantly for MtC gels (Table 2). This implies that the immobilization would not occur when the PMA is immobilized for the third time. The increase in the density of the PMA chains is interfering further immobilization process. This can be supplemented by SCA result. The phospholipid head groups on the surface of the collagen gel decreased in the SCA, implying that the surface of the CoPho gel was acquiring a hydrophilic nature (Fig. 2). The hydrophilicity of the CoPho gel was due to the MPC head group, which was located on the surface [18]. The hydrophilic nature of the MPC polymer is thought to be one factor that can suppress

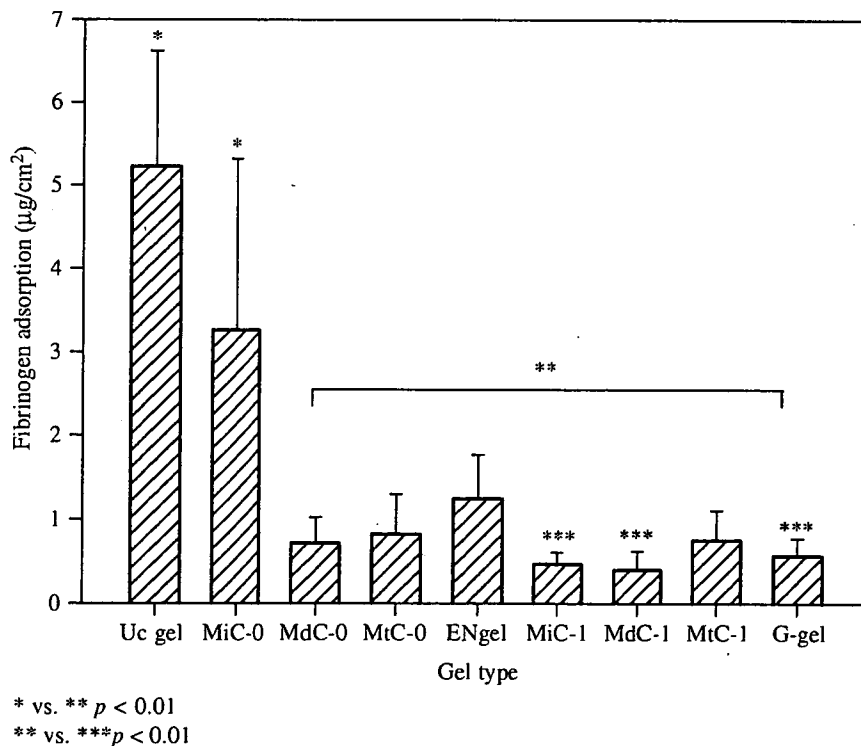


Fig. 6. Fibrinogen adsorption by the collagen gels. Each value represents the mean  $\pm$  SD ( $n = 5$ ).

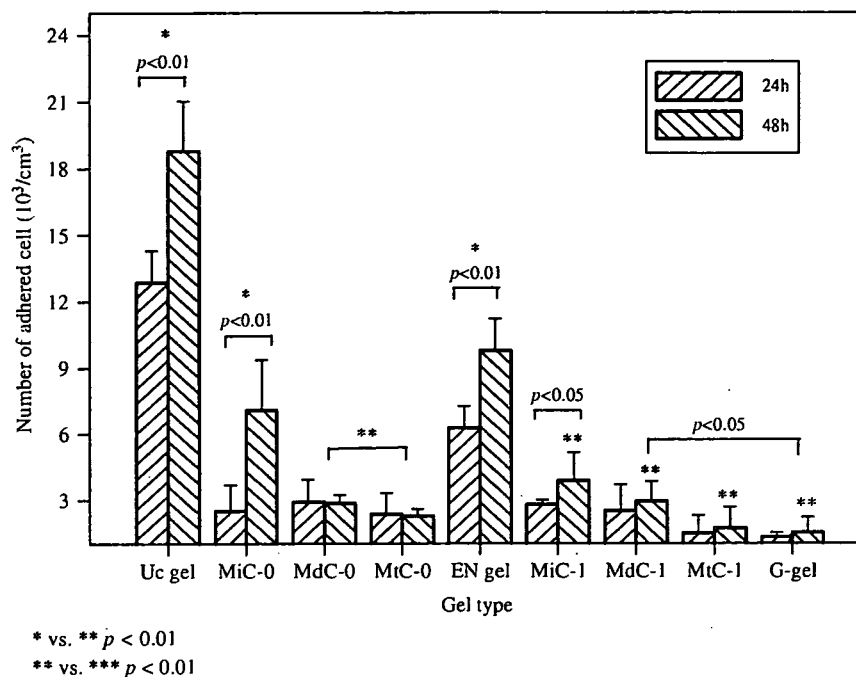


Fig. 7. Cell adhesion property of the respective collagen gels at a seeding density of 5000 cells/cm<sup>2</sup>. Each value represents the mean  $\pm$  SD ( $n = 5$ ).

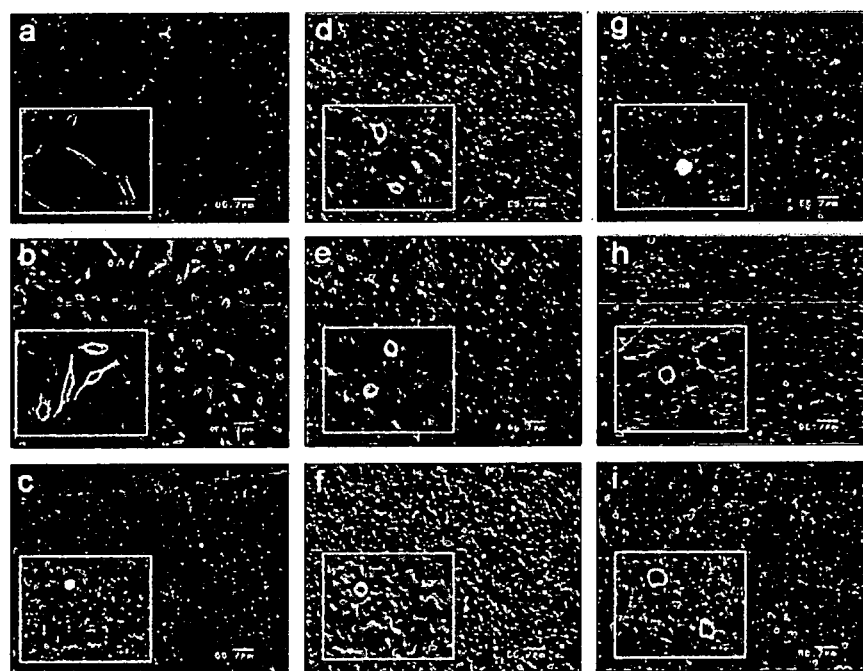


Fig. 8. SEM images of adhered fibroblast cells on respective collagen gels after 48 h of incubation. (a) Uc gel, (b) EN gel, (c) G-gel, (d) MiC-0, (e) MdC-0, (f) MtC-0, (g) MiC-1, (h) MdC-1, and (i) MtC-1 gel. SEM images in the large frame are shown at a magnification of  $\times 150$  and the small frame at a magnification of  $\times 1000$ .

the protein adsorption. It is because the wet condition of the surface is inducing the increase in the mobility of the MPC polymer head group. SCA further decreases as a result of the re-immobilization process, indicating an increase in the density of phospholipid head groups on the surface of the CoPho gels. However, third immobilization process did not decrease the contact angle further.

Did the increase in PMA on the surface of the collagen gel result in a change in the structure of the collagen gels as indicated in Fig. 1? We attempted to characterize the network structure by investigating the reacted amine group content and the swelling ratio (Figs. 3 and 4). Cross-linking collagen gels with EDC/NHS leads to a decrease in the number of reacted amine groups because  $\text{NH}_2$  from

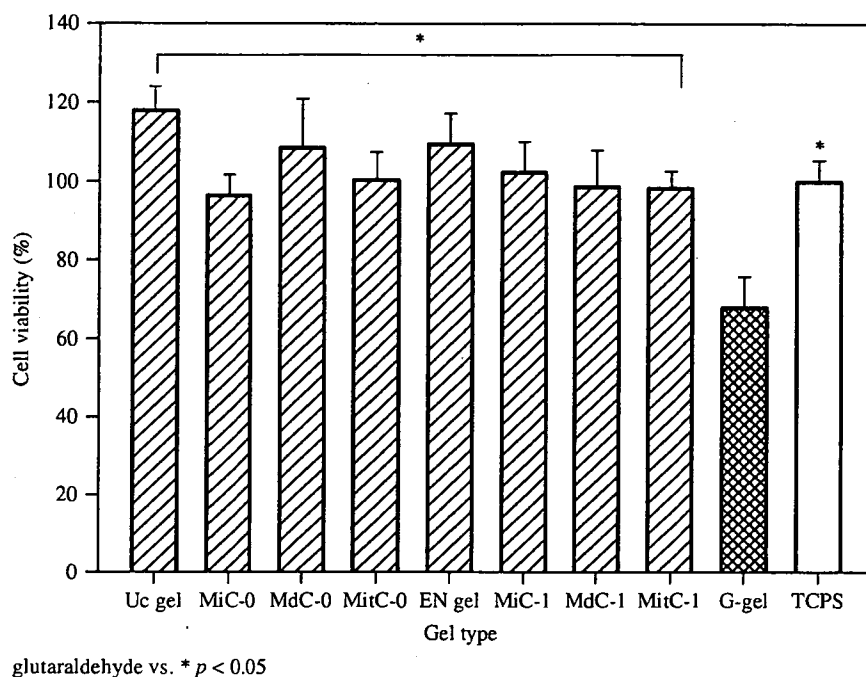


Fig. 9. Cell viability of L929 at 48 h of incubation with a seeding density of 5000 cells/cm<sup>2</sup> on the respective collagen gels. Each value represents the mean  $\pm$  SD ( $n = 3$ ).

(hydroxy-)lysine residues is consumed on amide bond formation and  $\text{NH}_3^+$  is not formed [7,21]. However, the percentage of the unreacted amine group content was higher than the expected number. Approximately, 60% of the  $\text{NH}_2$  remained unreacted compared to Uc gel. Immobilization of PMA on the collagen gel consumed approximately 40% of the amine groups. With regard to PMA immobilization, it is believed that the amine groups may be consumed only on the surface of the collagen gel because PMA cannot penetrate the collagen  $\alpha$ -helices [11]. The re-immobilization process decreases the reacted amine group content by up to 40% of Uc gel; immobilization process repeated three times, by up to 30% of Uc gel. The increase in the phosphorus concentration and the decrease in SCA were not observed, but the unreacted amine group content decreased for MtC gels. This implies that the immobilization is stopped, but few intra- and interhelical cross-links have occurred. This decrease is considered high when compared to that of G-gel, which demonstrates approximately 15% of Uc gel. Since the immobilization process occurs only on the surface of collagen gel, lowering the free amine content any further was not possible. An immobilization period of 48 h was the longest period of immobilization that showed a decrease in the number of unreacted amine group, and addition of a higher amount of EDC, NHS, and PMA during the cross-linking process did not cause a significant change in the reacted amine group content (data not shown).

The EN gel is formed by intra- and interhelical cross-links, whereas the CoPho gels are formed by a polymer-helix network. The formation of the cross-link network

leads to a decrease in the swelling ratio. In a previous study, we have mentioned that the swelling ratio of the collagen gels is expected to decrease with the progress in cross-linking [11]. The network formed by cross-linking would be dense, and this renders water absorption difficult for the gels. The swelling ratio under acidic and neutral pH conditions varies due to the repulsion force amongst the  $\text{NH}_3^+$  and  $\text{COO}^-$  groups. The swelling ratio is higher for collagen gels immersed in acidic pH condition because they stabilize under neutral pH conditions. With regard to collagen gels prepared with a 0.5 wt% collagen solution, highly acidic pH conditions causes the uncross-linked collagen gels to dissolve [11]; however, none of the collagen gels used in this study dissolved because the  $\alpha$ -helices were packed more tightly and were stabilized during gel preparation. Under neutral pH conditions, the collagen film would stabilize by forming a lattice network of fibrils comprising hydrophobic and electrostatic bonds [11,22–24]; hence, the swelling ratio would be less than that observed under acidic pH conditions. The difference in the swelling ratio between the collagen gels with and without inter- and intrahelical cross-links (under acidic pH conditions) is approximately 20–50%. The reacted amine group shows a difference of only 5–10% because stabilization of  $\alpha$ -helices by intra- and interhelical cross-links causes a decrease in the swelling ratio. The consumption of carboxyl groups and amine groups is eliminating the site for the protein binding. Arg-Gly-Asp (RGD) site is consumed for the immobilization process, which is making the protein more difficult to adsorbed [25,26]. This would be discussed in Section 4.2.

Formation of a denser network leads to difficulties in the degradation by collagenase (Fig. 5). Activation of collagenase requires adsorption on the collagen gel surface [11,27]. Subsequently, the collagenase penetrates the collagen gel and begins to cleave the helices [28]. However, a low swelling ratio does not permit collagenase adsorption by the CoPho gels. The degradation rate is considerably slower due to an increase in PMA density. A high density of phospholipid head groups is believed to prevent collagenase adsorption on the surface. Eventually, the CoPho gels would be more stable against collagenase.

#### 4.2. Biological properties of EN and CoPho gels

As previously mentioned, the hydrophilicity of the CoPho gel increases with immobilization of PMA due to the presence of phospholipid head group on the CoPho gel. High hydrophilicity is known to be one of the factors that lead to difficulties in protein adsorption [29]. Increase in the density of PMA results in a decrease in the adsorption rate of fibrinogen (Fig. 6). This implies that the immobilized MPC polymer leads to difficulties in the interaction of proteins with the gel surface. In addition to this, the cross-links also decrease protein adsorption. The  $\epsilon$ -amino groups from (hydroxy-)lysine residues of collagen are blocked by the cross-linking process [30]. Thus, it is believed that hydrophilicity of the hybrid gel and the blocked  $\epsilon$ -amino group renders it difficult for the hybrid gel surface to adsorb fibrinogen. A similar phenomenon was observed during the cell adhesion test (Fig. 7). We observed that repeated immobilization of PMA suppressed cell adhesion. Comparison of cell adhesion after 24- and 48-h cycles revealed that the number of adhered cells in the case of the Uc gel after 48 h cycle had increased by approximately 2 times; this rate of increase would decrease as collagen gels more phospholipids is immobilized. Immobilization of PMA did not induce any toxicity. Decrease in cell attachment on CoPho gels was entirely attributable to the surface property, i.e., the ability to regulate cell adhesion and protein adsorption. These results indicate that the immobilization of the PMA would induce almost the same effect as that of G-gel but without toxicity. As mentioned in Section 4.1, the formation of the cross-linking is eliminating the site for the protein binding. Same effect can be expected for the G-gels. Consumption of RGD for the cross-linking is making the gels to resist against protein adsorption and cell adhesion. Improper cross-linking by glutaraldehyde would induce the high protein adsorption [25]. However, in our case, G-gel showed low protein adsorption and cell adhesion, indicating that the RGD is effectively cross-linked. The formation of the cross-link is bringing the difficulty in the adsorption of proteins.

When assessing suppression of cell adhesion in terms of intra- and interhelical cross-links, we observed that suppression was considerably higher in gels with intra- and interhelical cross-links. The MtC-1 gel displayed cell

adhesivity that was similar to that of the G-gel. This implies that intra- and interhelical cross-links also constitute an important parameter in suppression of cell adhesion. This can be reaffirmed by the fact that the number of cells adhered onto the EN gel is less than that in the case of the Uc gel. Much higher effect can be seen for G-gel, but we could not detect high suppression of cell adhesion by EDC/NHS cross-link, indicating that the functional groups still exist largely on the surface. Comparing EN gel and MtC-0 gel, the reacted amine group content is almost the same but the biological property is different. This is due to the difference in the surface property of the EN gel and MtC-0 gel. Investigation of cell morphology revealed that the L929 cells were deformed on the non-MPC surface (Fig. 8). On the other hand, the cells remained intact (round) on the surface of the CoPho gels, indicating a weak interaction between cells and the surface [19]. However, the adsorption of protein and the adhesion of the cell were still higher compared to other materials that used MPC polymer [18,31–33]. Ishihara et al. pointed out that 30 mol% of MPC polymer is required for fibroblast suppression [29]. Repeated immobilization increased the number of phosphorylcholine moieties on the surface of collagen gel surface. However, it is believed that the increase in the number of phospholipids moieties is no longer possible, and no significant decrease was observed in the amount of adsorbed fibrinogen and adhered cells.

#### 5. Conclusion

Repeated immobilization of PMA can increase its immobilization rate, resulting in an increase in the number of MPC head groups; hence, unreacted amine group content and the swelling ratio decreased and the degradation by the collagenase was delayed. The cell morphology remained round indicating a weak interaction between the cells and the gel surface. Thus, the CoPho gel can be used as an alternative collagen-based gel for an implantable biomedical device. Furthermore, we expect that co-immobilization with different polymer-possessing carboxyl groups such as heparin is possible. In the near future, we look forward to reporting on the use of the CoPho gel *in vivo*.

#### Acknowledgments

This study was financially supported by a grant from the Research on Health Sciences focusing on Drug Innovation (KH61060) from the Japan Health Sciences Foundation and a grant from the Health and Labour Sciences Research Grants.

We would like to thank Dr. Kazuhiko Ishihara of The University of Tokyo for his kind assistance and advice on the preparation and analysis of the MPC polymer and Mr. James Sibarani of The University of Tokyo for his assistance on XPS analysis.

## References

- [1] Friess W. Collagen-biomaterials for drug delivery. *Eur J Pharm Biopharm* 1998;45:112–36.
- [2] Nimni ME, Cheung D, Strates B, Kodama M, Sheikh K. Chemically modified collagen: a natural biomaterial for tissue replacement. *J Biomed Mater Res* 1987;21:741–71.
- [3] Zeeman R, Dijkstra PJ, van Wachem PB, van Luyn MJ, Hendriks M, Cahalan PT, et al. Successive epoxy and carbodiimide cross-linking of dermal sheep collagen. *Biomaterials* 1999;20:921–31.
- [4] Yoshizato K, Nishikawa A, Taira T. Functionally polarized layers formed by epidermal cells on a permeable transparent collagen film. *J Cell Sci* 1988;91:491–9.
- [5] Weadock KS, Miller EJ, Bellincampi LD, Zawadsky JP, Dunn MG. Physical crosslinking of collagen fibers: comparison of ultraviolet irradiation and dehydrothermal treatment. *J Biomed Mater Res* 1995;29:1373–9.
- [6] Barbani N, Lazzeri L, Cristalli C, Cascone MG, Polacco G, Pizzirani G. Bioartificial materials based on blends of collagen and poly(acrylic acid). *J Appl Polym Sci* 1999;72:971–6.
- [7] Olde Damink LHH, Dijkstra PJ, van Luyn MJA, van Wachem PB, Nieuwenhuis P, Feijen J. Cross-linking of dermal sheep collagen using a water-soluble carbodiimide. *Biomaterials* 1996;17:765–73.
- [8] Wissink MJB, Beernink R, Pieper JS, Poot AA, Engbers GHM, Beugeling T, et al. Immobilization of heparin to EDC/NHS-cross-linked collagen. Characterization and in vitro evaluation. *Biomaterials* 2001;22:151–63.
- [9] van Luyn MJA, van Wachem PB, Olde Damink LHH. Relations between in vitro cytotoxicity and cross-linked dermal sheep collagens. *J Biomed Mater Res* 1992;26:1091–110.
- [10] Ishihara K, Nomura H, Mihara T, Kurita K, Iwasaki Y, Nakabayashi N. Why do phospholipid polymers reduce protein adsorption? *J Biomed Mater Res* 1998;39:323–30.
- [11] Nam K, Kimura T, Kishida A. Preparation and characterization of cross-linked collagen-phospholipid polymer hybrid gels. *Biomaterials* 2007;28:1–8.
- [12] Wissink MJB. Endothelialization of collagen matrices. Doctor's thesis, University of Twente, 1999; Chapter 4, p. 61–86.
- [13] Nam K, Watanabe J, Ishihara K. Characterization of the spontaneously forming hydrogels composed of water-soluble phospholipid polymers. *Biomacromolecule* 2002;3:100–5.
- [14] Olde Damink LH, Dijkstra PJ, Van Luyn MJ, Van Wachem PB, Nieuwenhuis P, Feijen J. Changes in the mechanical properties of dermal sheep collagen during in vitro degradation. *J Biomed Mater Res* 1995;29:139–47.
- [15] Bubnis WA, Ofner III. CM. The determination of  $\epsilon$ -amino groups in soluble and poorly soluble proteinaceous materials by a spectrophotometric method using trinitrobenzenesulfonic acid. *Anal Biochem* 1992;207:129–33.
- [16] Everaerts F, Torrianni M, van Luyn M, van Wachem P, Feijen J, Hendriks M. Reduced calcification of bioprostheses, cross-linked via an improved carbodiimide based method. *Biomaterials* 2005;25:5523–30.
- [17] Higuchi A, Sugiyama K, Yoon BO, Sakurai M, Hara M, Sumita M, et al. Serum protein adsorption and platelet adhesion on pluronic-adsorbed polysulfone membranes. *Biomaterials* 2003;24:3235–45.
- [18] Goda T, Konno T, Takai M, Moro T, Ishihara K. Biomimetic phosphorylcholine polymer grafting from polydimethylsiloxane surface using photo-induced polymerization. *Biomaterials* 2006;27:5151–60.
- [19] Watanabe J, Ishihara K. Phosphorylcholine and poly(D,L-lactic acid) containing copolymers as substrate for cell adhesion. *Artif Organs* 2003;27:242–8.
- [20] Lin Y, Wang L, Zhang P, Wang X, Chen X, Jing X, et al. Surface modification of poly(L-lactic acid) to improve its cytocompatibility via assembly of polyelectrolytes and gelatin. *Acta Biomater* 2006;2:155–64.
- [21] Khara AR, Peppas NA. Swelling/deswelling of anionic copolymer gels. *Biomaterials* 1995;16:559–67.
- [22] Ripamonti A, Roveri N, Briga D. Effects of pH and ionic strength on the structure of collagen fibrils. *Biopolymers* 1980;19:965–75.
- [23] Wallace D. The relative contribution of electrostatic interactions to stabilization of collagen fibrils. *Biopolymers* 1990;29:1015–26.
- [24] Rosenblatt J, Devereux B, Wallace D. Dynamic rheological studies of hydrophobic interactions in injectable collagen biomaterials. *J Appl Polym Sci* 1993;50:953–63.
- [25] Ber S, Köse T, Hasırcı V. Bone tissue engineering on patterned collagen films: and in vitro study. *Biomaterials* 2005;26:1977–86.
- [26] Chandy T, Das GS, Wilson RF, Rao GHR. Use of plasma glow for surface-engineering biomolecules to enhance blood compatibility of Dacron and PTFE vascular prosthesis. *Biomaterials* 2000;21:699–712.
- [27] Everaerts F, Torrianni M, van Luyn M, van Wachem P, Feijen J, Hendriks M. Reduced calcification of bioprostheses, cross-linked via an improved carbodiimide based method. *Biomaterials* 2005;25:5523–30.
- [28] Ma L, Gao C, Mao Z, Zhou J, Shen J. Enhanced biological stability of collagen porous scaffolds by using amino acids as novel cross-linking bridges. *Biomaterials* 2004;25:2997–3004.
- [29] Ishihara K, Ishikawa E, Iwasaki Y, Nakabayashi N. Inhibition of fibroblast cell adhesion on substrate by coating with 2-methacryloyloxyethyl phosphorylcholine polymers. *J Biomater Sci Polym Edn* 1999;10:1047–61.
- [30] Hersel U, Dahmen C, Kessler H. RGD modified polymers: biomaterials for stimulated cell adhesion and beyond. *Biomaterials* 2003;24:4385–415.
- [31] Yamasaki A, Imamura Y, Kurita K, Iwasaki Y, Nakabayashi N, Ishihara K. Surface mobility of polymers having phosphorylcholine groups connected with various bridging units and their protein adsorption-resistance properties. *Colloids Surf B: Biointerf* 2003;28:53–62.
- [32] Watanabe J, Ishihara K. Cell engineering biointerface focusing on cytocompatibility using phospholipid polymer with an isomeric oligo(lactic acid) segment. *Biomacromolecules* 2005;6:1797–802.
- [33] Goda T, Konno T, Takai M, Moro T, Ishihara K. Photoinduced phospholipid polymer grafting on Parylene film: advanced lubrication and antibiofouling properties. *Colloids Surf B: Biointerfaces* 2007;54:67–73.

ORIGINAL ARTICLE

Tsuyoshi Kimura, PhD · Sayaka Iwai  
Toshiyuki Moritan, PhD · Kwangwoo Nam, PhD  
Shingo Mutsuo · Hidekazu Yoshizawa, PhD  
Masahiro Okada, PhD · Tsutomu Furuzono, PhD  
Tosihya Fujisato, PhD · Akio Kishida, PhD

## Preparation of poly(vinyl alcohol)/DNA hydrogels via hydrogen bonds formed on ultra-high pressurization and controlled release of DNA from the hydrogels for gene delivery

**Abstract** Poly(vinyl alcohol) (PVA) hydrogels interacting with DNA mediated by hydrogen bonds (PVA/DNA hydrogel) were developed using ultra-high pressure (UHP) technology. The goal was to create a new method of gene delivery by controlled release of DNA. Mixed solutions of DNA and PVA at various concentrations were pressurized at 10 000 atmospheres at 37°C for 10 min. PVA/DNA hydrogels with good formability were produced at PVA concentrations of more than 5% w/v. The presence of DNA in the obtained hydrogels was confirmed by spectroscopic analysis and nucleic acid dye staining. DNA release from the hydrogels was investigated using PVA/DNA hydrogel samples of 5% and 10% w/v formed by UHP treatment or by conventional freeze–thaw methods. The DNA release curves from both types of samples showed a rapid phase in the initial 15 h followed by a sustained release phase. However, there was a difference in the amount of DNA released. Less DNA was released by the pressurized hydrogels than by the freeze–thaw hydrogels. Also, the cumulative amount of DNA released decreased as the PVA content in the hydrogels increased. These results indicate that DNA release from the hydrogels can be modulated by changing

the preparation method and the PVA content. Furthermore, it was demonstrated that DNA release could be controlled by varying the amount and duration of pressurizing used to form the hydrogels. Intact fractions of plasmid DNA released from the hydrogels were separated by agarose gel electrophoretic analysis. These results suggest that, using controlled release, DNA from PVA/DNA hydrogels formed by UHP treatment can be transfected into cells.

**Key words** Controlled release · Ultra-high pressure · DNA · Hydrogel · Poly(vinyl alcohol)

Received: March 31, 2006 / Accepted: November 18, 2006

T. Kimura · K. Nam · A. Kishida (✉)  
Institute of Biomaterials and Bioengineering, Tokyo Medical and Dental University, 2-3-10 Kanda-Surugadai, Chiyoda-ku, Tokyo 101-0062, Japan  
Tel. and Fax +81-3-5280-8028  
e-mail: kishida.fm@tmd.ac.jp

S. Iwai · T. Moritan  
Department of Medical Engineering, Suzuka University of Medical Science, Suzuka, Japan

S. Mutsuo · H. Yoshizawa  
Department of Environmental Chemistry and Materials, Okayama University, Okayama, Japan

M. Okada · T. Furuzono  
Department of Biomedical Engineering, National Cardiovascular Center Research Institute, Osaka, Japan

T. Fujisato  
Department of Regenerative Medicine and Tissue Engineering, National Cardiovascular Center Research Institute, Osaka, Japan

### Introduction

Safe and biocompatible synthetic materials have been developed as biomaterials.<sup>1</sup> In gene therapy, nonviral synthetic gene carriers have been the focus of attention due to their biological safety advantages over viruses.<sup>2</sup> In many cases, cationic synthetic materials, such as cationic lipids, liposomes,<sup>3</sup> polyethyleneimine,<sup>4</sup> polyamideamine dendrimer,<sup>5</sup> poly-L-lysine (PLL), PLL derivatives,<sup>6</sup> and other cationic peptides,<sup>7</sup> have been used as nonviral vectors. It is possible to form complexes between these materials and DNA using the electrostatic interaction between their cationic groups and the anionic groups of DNA, making the DNA robust against nuclease degradation and enabling effective transfection into mammalian cells.<sup>8,9</sup> However, the cytotoxicity of cationic materials was reported to be a significant problem.<sup>10,11</sup> For safer and more efficient gene delivery, it is necessary to develop a noncationic or less cationic gene carrier through nonelectrostatic interaction with DNA. Sakurai et al. reported that a triple helical complex of single-strand DNA and double-strand schizophyllan, which is a kind of polysaccharide ( $\beta$ -1,3 glucan), was formed through hydrogen bonding.<sup>12</sup> In addition, we previously reported that nanoparticles of poly(vinyl alcohol) (PVA) bonded to DNA via hydrogen bonds were obtained when mixed solutions of PVA (less than 0.01% w/v) and DNA were treated under ultra-high pressure (UHP) at



10 000 atmospheres (980 MPa) and 40°C for 10 min.<sup>13</sup> It is well known that intra- and intermolecular hydrogen bonding increases in these conditions.<sup>14</sup> The PVA/DNA nanoparticles could be internalized into mammalian cells, suggesting that they have utility as a novel nonviral vector that uses nonelectronic interactions.

Recently, controlled release of DNA was also investigated as a possible method of enhancing transfection efficiency using various biomaterials such as poly (lactide-co-glycolide) (PLGA),<sup>15</sup> hyaluronic acid,<sup>16</sup> atelocollagen,<sup>17</sup> and gelatin.<sup>18,19</sup> Shea et al. reported that the sustained delivery of DNA from PLGA led to effective transfection of a large number of cells *in vitro* and *in vivo*.<sup>15</sup> However, it was difficult to regulate the release of DNA owing to the lack of interaction forces, such as covalent, electrostatic, and hydrogen bonding, with which DNA molecules are loaded into PLGA with polymer molecules. Tabata et al. reported enhancement and prolongation of gene expression using a cationized gelatin hydrogel interacting with DNA electrostatically.<sup>18,19</sup> The controlled release of DNA depended on hydrogel degradation, but the cationized gelatin hydrogel was crosslinked by glutaraldehyde, which has generally cytotoxic properties, to obtain different degrees of cationization.

In the present study, we report the preparation of a novel PVA hydrogel with DNA crosslinked physically by hydrogen bonds using UHP technology and its application to the controlled release of DNA. The goal is to develop an effective, low-cytotoxic and gene-releasable biomaterial. PVA/DNA hydrogels were obtained for various pressurization conditions, temperatures, and processing times. DNA release from the hydrogels was investigated *in vitro*. PVA is widely used for biomedical applications because of its biocompatibility and neutrally charged nature.<sup>20</sup> It is also known that PVA hydrogel is formed by physical crosslinking with hydrogen bonds when PVA solution is frozen and thawed several times, which is called the freeze-thaw method.<sup>21</sup>

## Materials and methods

### Materials

In our experiments, we used PVA samples with an average molecular weight of 74 800 and a degree of saponification of 99.8%, as supplied by Kuraray (Osaka, Japan). We also used salmon sperm DNA purchased from Wako (Osaka, Japan), plasmid DNA encoding enhanced green fluorescence protein under a cytomegalovirus promoter (pEGFP-N1, BD Science, Palo Alto, CA, USA), and nucleic acid staining dye solution (Mupid Blue) obtained from Advance (Tokyo, Japan).

### Preparation of PVA/DNA hydrogels by UHP

Aqueous PVA solutions of 6%, 8%, 10%, 14%, and 20% w/v were prepared by autoclaving three times for 30 min at

121°C. Salmon sperm DNA was dissolved in a Tris-EDTA buffer (TE, pH = 7.8) at a concentration of 16.3 mg/ml. The DNA solution was mixed with PVA solutions of 10%, 14%, and 20% w/v at a ratio of 1:1. The 0.7-ml samples were transferred in silicon tubes (9 × 25 mm) with both ends capped by silicon plugs. The tubes were pressurized under various UHP conditions, using different pressures, temperatures, and durations, in a high-pressure machine (Kobe Steel, Kobe, Japan).

### Confirmation of the presence of DNA in the PVA/DNA hydrogels

The presence of DNA in the PVA/DNA hydrogels produced by UHP treatment was confirmed by nucleic acid dye staining and UV-visible spectroscopy. For the former method, the PVA/DNA hydrogels were immersed in nucleic acid dye solution for 1 min and then transferred to 70% ethanol. After 1 min, they were immersed in ion-exchanged water for 1 min. For the latter method, after the PVA/DNA hydrogels were melted at 90°C for 10 min, their DNA concentration was measured by a spectrophotometer (V-560, JASC, Tokyo, Japan).

### DNA release from hydrogels

The PVA/DNA hydrogels prepared by UHP were immersed in 5 ml of phosphate-buffered saline (PBS) for 144 h at 37°C. At 0.25, 0.5, 2, 3, 15, 27, 48, 111, and 144 h, 20 µl of the samples in the outer part of the PBS solution was collected and the DNA concentration was measured spectrophotometrically at 260 nm (Gene Quant Pro S, Amersham, Tokyo, Japan).

### Stability of plasmid DNA released from hydrogels

Plasmid DNA (pDNA) was used instead of salmon sperm DNA and the mixed solutions of pDNA (100 µg/ml) and PVA (5% or 10% w/v) were treated by UHP under the conditions described above. The obtained PVA/pDNA hydrogels were immersed in PBS for 12 and 48 h, and then the samples in the outer part of the solution were collected and analyzed by agarose gel electrophoresis at 100 V for 45 min.

## Results and discussion

Aqueous solutions of PVA at concentrations ranging from 3% to 10% w/v were hydrostatically pressurized at 10 000 atm at 37°C for 10 min. With a PVA solution of 3% w/v, the clear solution was transformed into a turbid and viscous solution by pressurization (Fig. 1A). An aggregation of PVA particles with an average diameter of 1 µm was observed in the PVA solution on scanning electron microscopy (SEM, data not shown). For PVA concentrations of more than 4% w/v, hydrogels were produced on pressuriza-

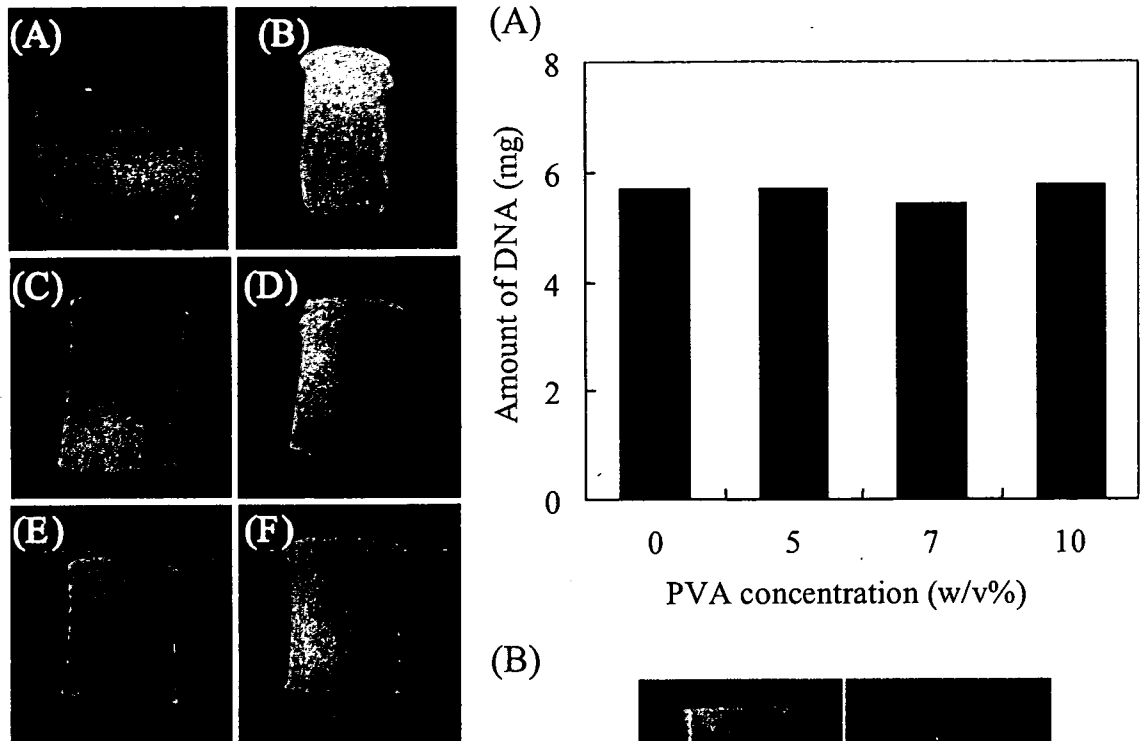


Fig. 1. Photographs of poly(vinyl alcohol) (PVA) hydrogels (A–D) and PVA/DNA (E,F) hydrogels at concentrations of A 3% w/v, B 4% w/v, C,E 5% w/v, and D,F 10% w/v obtained by ultra-high pressure treatment.

tion (Fig. 1B–D). The PVA hydrogel of 4% w/v was fragile (Fig. 1B), but increasing the PVA concentration enhanced hydrogel formability, and hard hydrogels were obtained at a PVA concentration of 10% w/v (Fig. 1D). These results indicate that pressurization induced physical cross-linking of PVA molecules and that the degree of cross-linking increased as the PVA concentration increased. To investigate whether the PVA molecules were physically cross-linked by hydrogen bonding, a PVA solution of 5% w/v with urea (3.3M), which was used as a hydrogen bond inhibitor, was treated under the above pressurizing conditions. The solution remained translucent (data not shown), indicating that the PVA hydrogel obtained by pressurization was mediated by hydrogen bonding.

The gelation of mixed solutions of DNA and PVA (5% and 10% w/v) was achieved by pressurization in the conditions described above (Fig. 1E,F). To confirm the presence of DNA in the hydrogels obtained, they were heat treated at 90°C for 10min and then the DNA concentration of the solutions obtained was measured spectrophotometrically at 260nm. Roughly equal amounts of DNA were contained in each hydrogel (Fig. 2A). Also, when the hydrogels were immersed in nucleic acid dye solution, which interacts electrostatically with the phosphate groups of DNA, the PVA hydrogel with DNA was stained, whereas the PVA hydrogel without DNA was not (Fig. 2B). These results indicate that a PVA hydrogel that sustains DNA (PVA/DNA hydrogel) was formed on pressurization. On the other hand,

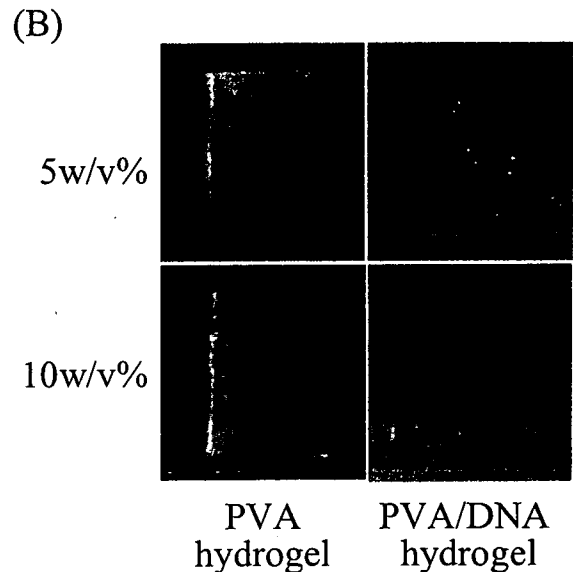
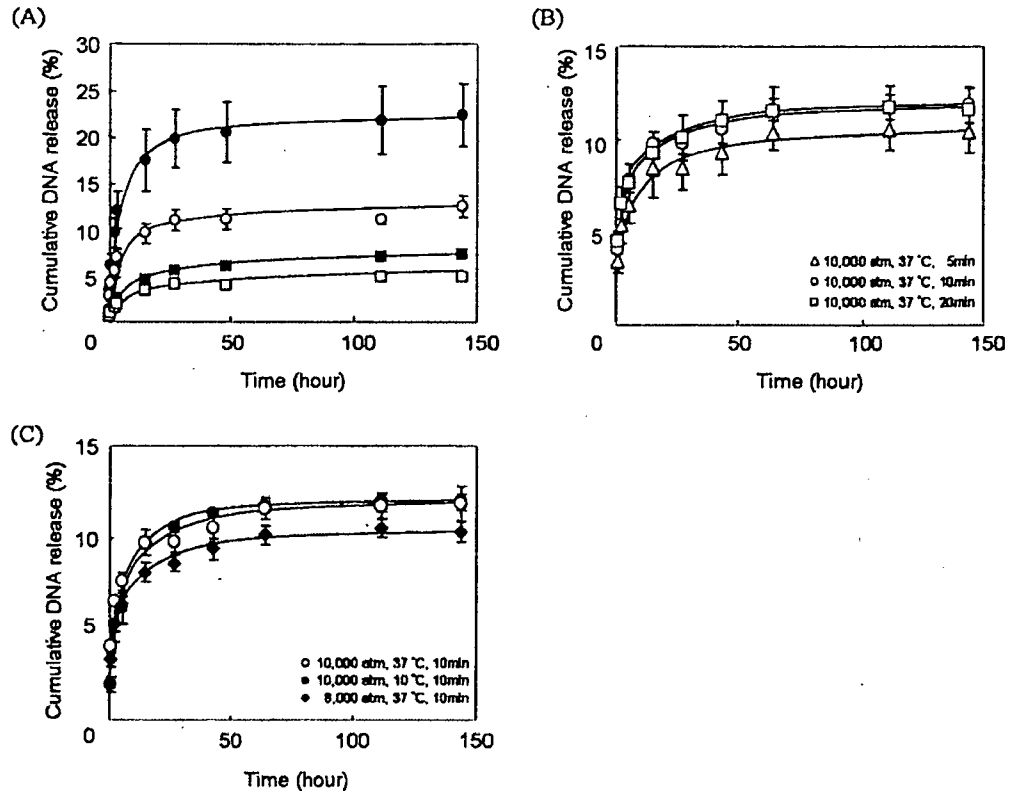


Fig. 2A,B. Presence of DNA in PVA/DNA hydrogels. A Amount of DNA in solution obtained by melting PVA/DNA hydrogels prepared using ultra-high pressure processing. B Photographs of PVA hydrogels and PVA/DNA hydrogels stained with nucleic acid dye

when urea was introduced, PVA/DNA hydrogel was not obtained on pressure treatment. This result suggests that hydrogen bonding between PVA and DNA took place in the pressurized PVA/DNA hydrogel.

DNA release from the PVA/DNA hydrogel formed by pressurization at 10000atm at 37°C for 10min was investigated. PVA/DNA hydrogels produced by the freeze-thaw method, a common method of forming PVA hydrogels,<sup>21</sup> were used as control samples. Figure 3A shows DNA release profiles from the PVA/DNA hydrogels at PVA concentrations of 5% and 10% w/v obtained by pressurization and the freeze-thaw method. Each release curve of DNA from a hydrogel consisted of a rapid phase in the initial 15h followed by a sustained release phase. However, the amount

**Fig. 3A–C.** DNA release test from PVA/DNA hydrogels produced by pressurization under various conditions or by the freeze–thaw method. **A** Release profiles of DNA from hydrogels at PVA concentrations of 5% w/v (○, ●) and 10% w/v (□, ■) PVA concentration. *Open* and *solid* symbols indicate DNA from hydrogels obtained by pressurization (at 10000 atm and 37°C, 10min) and the freeze–thaw method, respectively. **B** Release profiles of DNA from hydrogels of 5% w/v obtained by pressurization at 10000 atm and 37°C for 5 min (□), 10 min (○), and 20 min (△). **C** Release profiles of DNA from hydrogels of 5% w/v obtained by pressurization at 10000 atm and 37°C (○), 10000 atm and 10°C (●), and 8000 atm and 37°C (□) for 10 min

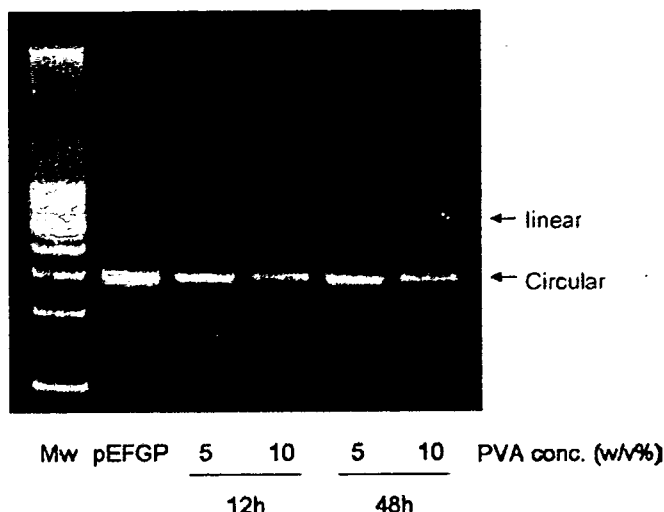


of DNA released was dependent on PVA content and on which procedure was used to prepare the hydrogels. The DNA release from the 10% w/v PVA/DNA hydrogels was lower than that from the 5% w/v PVA/DNA hydrogels, irrespective of the preparation methods. This is consistent with the fact that the 5% w/v samples were more easily stained by nucleic acid dye than the 10% w/v samples. We suppose that the increased crosslinking in the hydrogel caused by the increase in the PVA content contributed to the reduction of DNA released from the hydrogel. On the other hand, at the same PVA concentrations, DNA was more effectively released from the freeze–thaw hydrogels than from the pressurized hydrogels. Fibrous structures with large spaces (larger than 1  $\mu\text{m}$ ) were observed on SEM in the hydrogels made from 5% w/v PVA obtained by the freeze–thaw method, while many porous structures with diameters of 300  $\mu\text{m}$  were observed in the pressurized hydrogels (data not shown). We believe that this difference in internal structure between sample types affected the interaction of PVA and DNA, resulting in the larger release of DNA from the freeze–thaw hydrogels.

To investigate the influence of the pressure conditions used to form hydrogels on DNA release, PVA/DNA hydrogels of 5% w/v were prepared by different levels of pressurization at different temperatures and for different durations. First, with pressure processing periods varying from 5 to 20 min at 10000 atm and 37°C, similar DNA release profiles were exhibited for the hydrogels obtained at pressurizing times of 10 and 20 min, but the amount of DNA released by hydrogel samples pressurized for 5 min (Fig. 3B) was less than that released by samples with longer pres-

surizing times. Second, the DNA release curves of the PVA/DNA hydrogel produced on pressurization at 10000 atm and 10°C for 10 min were the same as those for hydrogels produced on pressurization at 10000 atm and 37°C for 10 min. However, less DNA was released by hydrogels produced at pressures of 8000 atm and 37°C for 10 min than by hydrogels produced at 10000 atm and 37°C for 10 min (Fig. 3C). These results indicate that DNA release from pressurized hydrogels is dependent on the level and duration of pressure used in the hydrogel formation process. We previously reported that PVA gelation was promoted by increasing the pressure and by prolonging the pressurization time, by which close hydrogen bonds between PVA molecules are formed.<sup>22</sup> It seems that DNA was easily released from PVA/DNA hydrogels pressurized under conditions of more than 10000 atm for longer than 10 min because the hydrogen bonding interaction between PVA and DNA was more unstable than that between PVA molecules under more intense pressure conditions.

It is important for DNA to be released from hydrogels without structural change or degradation.<sup>2,23</sup> Plasmid DNA (pDNA), which is generally used as the DNA delivered by a nonviral vector, was used instead of salmon sperm DNA. PVA/pDNA hydrogels at PVA concentrations of 5% and 10% w/v were obtained by pressurization at 10000 atm at 37°C for 10 min and then immersed in 5 ml PBS. After 12 and 48 h of immersion, the outer part of the solution was collected and analyzed by agarose gel electrophoresis at 100 V for 30 min to investigate the stability of released pDNA from the hydrogels (Fig. 4). No degradation of DNA was observed, indicating that the plasmid DNA released



**Fig. 4.** Agarose gel electrophoresis of plasmid DNA (pDNA) released from PVA/pDNA hydrogels with PVA concentrations of 5% and 10% w/v produced by pressurization at 10000 atm and 37°C for 10 min after immersion in phosphate-buffered saline for 12 and 48 h

from the PVA/DNA hydrogels was stable. Two bands of linear and circular plasmid DNA were observed with 5% w/v PVA/DNA hydrogel, while circular plasmid DNA was released from the 10% w/v PVA/DNA hydrogel, indicating that the linear form of plasmid DNA tends to interact more strongly with PVA than the circular plasmid DNA.

## Conclusions

Novel PVA/DNA hydrogels crosslinked physically by hydrogen bonds were developed using UHP technology. DNA released from the hydrogels was controlled by varying the PVA concentration and pressurization conditions, such as the level and duration of pressure used to form the hydrogels. The demonstrated stability of the DNA released from the hydrogels suggests that PVA/DNA hydrogels have potential as a candidate for gene delivery.

**Acknowledgments** This work was supported by grants from the Ministry of Health, Labor and Welfare, of Japan and the Ministry of Education, Culture, Sports, Science and Technology of Japan. We thank Kuraray, Co., Ltd., for supplying the poly(vinyl alcohol).

## References

- Nowak T, Nishida K, Shimoda S, Konno Y, Ichinose K, Sakakibara M, Shighiri M, Nakabayashi N, Ishihara K. Biocompatibility of MPC: in vivo evaluation for clinical application. *J Artif Organs* 2000;1:39-46
- Glover DJ, Lipps HJ, Jans DA. Towards safe, non-viral therapeutic gene expression in humans. *Nat Rev Genet* 2005;6:299-310

- Zhang S, Xu Y, Wan B, Qiao W, Liu D, Li Z. Cationic compounds used in lipoplexes and polyplexes for gene delivery. *J Control Release* 2004;100:165-180
- Lungwitz U, Breunig M, Blunk T, Göpferich A. Polyethylenimine-based non-viral gene delivery systems. *Eur J Pharm Biopharm* 2005;60:247-266
- Dufes C, Uchegbu IF, Scatzlein AG. Dendrimers in gene delivery. *Adv Drug Deliv Rev* 2005;57:2117-2202
- Kimura T, Yamaoka T, Iwase R, Murakami A. Effect of physico-chemical properties of polyplexes composed of chemically modified PL derivatives on transfection efficiency in vitro. *Macromol Biosci* 2002;2:437-446
- Futaki S. Membrane-permeable arginine-rich peptides and the translocation mechanisms. *Adv Drug Deliv Rev* 2005;57:547-558
- Reschel T, Koňák C, Oupický D, Seymour LW, Ulbrich K. Physical properties and in vitro transfection efficiency of gene delivery vectors based on complexes of DNA with synthetic polycations. *J Control Release* 2002;81:201-217
- Elouahabi A, Ruyschaert JM. Formation and intracellular trafficking of lipoplexes and polyplexes. *Mol Ther* 2005;11:336-347
- Fischer D, Li Y, Ahlemeyer B, Krieglstein J, Kissel T. In vitro cytotoxicity testing of polycations: influence of polymer structure on cell viability and hemolysis. *Biomaterials* 2006;24:1121-1131
- Choksalunimitr S, Matsuda S, Tokuda H, Takakura Y, Hashida M. In vitro cytotoxicity of macromolecules in different cell culture systems. *J Control Release* 1995;34:233-241
- Sakurai K, Mizu M, Shinkai S. Polysaccharide-polynucleotide complexes. 2. Complementary polynucleotide mimic behavior of the natural polysaccharide schizophyllan in the macromolecular complex with single-stranded RNA and DNA. *Biomacromolecules* 2001;2:641-650
- Kimura T, Okuno A, Miyazaki K, Furuzono T, Ohya Y, Ouchi T, Mutsuo S, Yoshizawa H, Kitamura Y, Fujisato T, Kishida A. Novel PVA-DNA nonparticles prepared by ultra high pressure technology for gene delivery. *Mater Sci Eng C* 2004;24:797-801
- Doi E, Shimizu A, Kitabatake N. Gel-sol transition of ovalbumin by high pressure. In: Hayashi R (ed) *High pressure bioscience and food science*. Kyoto: Sanei Press, 1993;171-177
- Shea LD, Smiley E, Bonadio J, Mooney DJ. DNA delivery from polymer matrices for tissue engineering. *Nat Biotech* 1999;17:551-554
- Chun KW, Lee JB, Kim SH, Rark TG. Controlled release of plasmid DNA from photo-cross-linked pluronic hydrogels. *Biomaterials* 2005;26:3319-3326
- Ochiya T, Takahama Y, Nagahara S, Sumita Y, Hisada A, Itoh H, Nagai Y, Terada M. New delivery system for plasmid DNA in vivo using atelocollagen as a carrier material: the Minipellet. *Nat Med* 1999;5:707-710
- Fukunaka Y, Iwanaga K, Morimoto K, Kakemi M, Tabata Y. Controlled release of plasmid DNA from cationized gelatin hydrogels based on hydrogel degradation. *Biomaterials* 2005;26:3319-3326
- Kushibiki T, Tomoshige R, Fukunaka Y, Kakemi M, Tabata Y. In vivo release and gene expression of plasmid DNA by hydrogels of gelatin with different cationization extents. *J Control Release* 2003;90:207-216
- Miyashita H, Shimmura S, Kobayashi H, Taguchi T, Asano-Kato K, Uchino Y, Kato M, Shimazaki J, Tanaka J, Tsubota K. Collagen-immobilized poly (vinyl alcohol) as an artificial cornea scaffold that supports a stratified corneal epithelium. *J Biomed Mater Res Part B: Appl Biomater* 2006;76B:56-63
- Hyon SH, Cha WI, Ikada Y. Preparation of transparent poly (vinyl alcohol) hydrogel. *Polymer Bull* 1989;22:119-122
- Yamamoto K, Furuzono T, Kishida A, Mutsuo S, Yoshizawa H, Kitamura Y. Formation of a supramolecular assembly of poly (vinyl alcohol) by ultrahigh pressure. Meeting Report of the Poval Committee 2002;121:25-26
- Walter E, Moelling K, Pavlovich HP. Microencapsulation of DNA using poly (D,L-lactide-co-glycolide): stability issues and release characteristics. *J Control Release* 1999;61:361-374